

Electrical and magnetic properties of ion-exchangeable layered ruthenates

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Abstract

An ion-exchangeable ruthenate with a layered structure, $K_{0.2}RuO_{2.1}$, was prepared by solid-state reactions. The interlayer cation was exchanged with H^+ , $C_2H_5NH_3^+$, and $((C_4H_9)_4N^+)$ through proton-exchange, ion-exchange, and guest-exchange reactions. The electrical and magnetic properties of the products were characterized by DC resistivity and susceptibility measurements. Layered $K_{0.2}RuO_{2.1}$ exhibited metallic conduction between 300 and 13 K. The products exhibited similar magnetic behavior despite the differences in the type of interlayer cation, suggesting that the ruthenate sheet in the protonated form and the intercalation compounds possess metallic nature.

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1. Introduction

The capability of ion-exchangeable layered oxides to accommodate various organic cations and molecules in the two-dimensional interlayer space makes such material an interesting target for ordered inorganic–organic hybrid materials with mixed characteristic of the inorganic oxide layer and the organic interlayer [1–4]. In general, the inorganic layer is composed of an insulating or poorly conducting oxide layer such as titanates, niobates, and tantalates [1–4]. Some examples of carrier doping into ion-exchangeable layered transition metal oxides have been reported, e.g., Mo- and W-based bronze, layered vanadium oxides (see Refs. [5–7] for comprehensive reviews), and more recently, Na_xCoO_2 [8]. Such materials are conductive when the transition metal is in a mixed valence state, i.e., in the

doped state, thus the conductivity is sensitive to the amount of interlayer cation.

Ruthenium dioxide, RuO_2 , is a unique oxide that has metallic conduction and can be easily handled under ambient conditions [9,10]. In addition, it is photochemically and electrochemically active and stable in acidic and alkaline electrolytes [9,10]. Due to such characteristics, RuO_2 has been widely used as catalysts, electrocatalysts, bottom electrodes, and electrode materials for electrochemical capacitors (also known as supercapacitors, ultracapacitors, or pseudocapacitors) [10–16].

Ruthenates with layered structures, such as Li_xRuO_2 [17–19] and Sr_2RuO_4 [20] exhibit metallic behavior. However, these compounds do not have ion-exchange capabilities; thus chemical modification of these layered ruthenates is intricate. Quite recently, two new layered ruthenates have been synthesized, $K_{0.2}RuO_{2.1} \cdot nH_2O$ [21,22] and $Na_xRuO_2 \cdot nH_2O$ [23]. The interlayer potassium ions in $K_{0.2}RuO_{2.1} \cdot nH_2O$ could be exchanged to protons to give the protonic form, $H_{0.2}RuO_{2.1} \cdot nH_2O$, by a simple proton-exchange reaction in acid [22].

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Furthermore, organic cations such as ethylammonium and tetrabutylammonium could be intercalated into the interlayer space. In addition, delamination (exfoliation) into ruthenic acid nanosheets could be accomplished by dispersing the tetrabutylammonium-ruthenic acid intercalation compound in solvents [22]. Such material could be used as electrode materials for electrochemical capacitors, due to the high electrochemically active surface area and presumable high conductivity [22,24]. Here we report the electrical and magnetic properties of layered $\text{K}_{0.2}\text{RuO}_{2.1}$ and the magnetic properties of the derivatives with various interlayer cations, namely, $\text{H}_{0.2}\text{RuO}_{2.1}$, $(\text{C}_2\text{H}_5\text{NH}_3)_{0.2}\text{RuO}_{2.1}$ and $((\text{C}_4\text{H}_9)_4\text{N})_{0.2}\text{RuO}_{2.1}$.

2. Experimental

Layered $\text{K}_{0.2}\text{RuO}_{2.1}$ was obtained by a solid-state reaction of a pelletized mixture of K_2CO_3 and RuO_2 (5:8 molar ratio) under Ar flow at 850°C for 12 h. The product was ground into fine powder and then washed with copious amounts of water and dried at 120°C . Layered $\text{K}_{0.2}\text{RuO}_{2.1}$ was converted into the protonic form, layered $\text{H}_{0.2}\text{RuO}_{2.1} \cdot n\text{H}_2\text{O}$, by acid treatment with 0.5 M H_2SO_4 at 60°C for 48 h, followed by washing with copious amounts of water. $\text{H}_{0.2}\text{RuO}_{2.1} \cdot n\text{H}_2\text{O}$ was dried at 120°C to remove interlayer water before characterization of the structure, electric and magnetic properties. The cation ratio of the product after the acid treatment was $\text{K}:\text{Ru} = <0.05:1$, based on ICP, EDAX and XPS analysis [22]. The ethylammonium-ruthenic acid intercalation compound, $(\text{C}_2\text{H}_5\text{NH}_3)_{0.2}\text{RuO}_{2.1}$, was prepared by reaction of layered $\text{H}_{0.2}\text{RuO}_{2.1} \cdot n\text{H}_2\text{O}$ with a 50% ethylamine aqueous solution. The solid product was centrifugally collected (15,000 rpm), washed with acetone and vacuum dried. The tetrabutylammonium-ruthenic acid intercalation compound, $((\text{C}_4\text{H}_9)_4\text{N})_{0.2}\text{RuO}_{2.1}$, was prepared by reaction of the ethylammonium-ruthenic acid intercalation compound with a 10% tetrabutylammonium hydroxide aqueous solution. The solid product was centrifugally collected (15,000 rpm). The X-ray diffraction patterns of the products were in agreement with our previous study [22]. The basal spacings measured from the lowest angle reflections for $\text{K}_{0.2}\text{RuO}_{2.1}$, $\text{H}_{0.2}\text{RuO}_{2.1}$, $(\text{C}_2\text{H}_5\text{NH}_3)_{0.2}\text{RuO}_{2.1}$, $((\text{C}_4\text{H}_9)_4\text{N})_{0.2}\text{RuO}_{2.1}$, were 0.729, 0.456, 0.815, 1.690 nm, respectively. The difference in the basal spacing is in reasonable agreement with the size of the interlayer cations.

Resistivity data were collected from 280 to 13 K using a standard DC four-probe method with Ag paste contacts. Polycrystalline powders of $\text{K}_{0.2}\text{RuO}_{2.1}$ and rutile-type RuO_2 were pressed into pellets and sintered again at 850°C for 12 h under Ar flow for the resistivity measurements. Rutile-type RuO_2 was obtained by

calcination of commercial $\text{RuO}_2 \cdot n\text{H}_2\text{O}$ (Johnson Matthey) at 850°C for 12 h under Ar flow. Since layered $\text{H}_{0.2}\text{RuO}_{2.1}$ undergoes interlayer dehydration and condensation at elevated temperatures, the resistivity data was collected using a non-sintered pellet of $\text{H}_{0.2}\text{RuO}_{2.1}$ dried at 120°C . Resistivity measurements for a non-sintered pellet of amorphous $\text{RuO}_2 \cdot n\text{H}_2\text{O}$ ($n \sim 0.8$) dried at 150°C were also conducted, since this composition has been reported as to be the most suitable for electrochemical capacitors [15,16]. Magnetic susceptibility of powder samples was measured in a magnetic field of 1 T from 300 to 5 K using a superconducting quantum interference device (SQUID) magnetometer (MPMS-7, Quantum Design) at Waseda University Material Characterization Central Laboratory. All of the analysis was conducted on anhydrous materials except for $\text{RuO}_2 \cdot n\text{H}_2\text{O}$.

3. Results and discussion

The normalized resistivity of layered $\text{K}_{0.2}\text{RuO}_{2.1}$ is compared with rutile-type RuO_2 in Fig. 1. The room-temperature resistivity of layered $\text{K}_{0.2}\text{RuO}_{2.1}$ was $2\text{ m}\Omega\text{ cm}$, which is comparable to that of rutile-type RuO_2 ($0.5\text{ m}\Omega\text{ cm}$). The temperature dependency of the resistivity for layered $\text{K}_{0.2}\text{RuO}_{2.1}$ slightly increased with increasing temperature and exhibited a similar tendency to that of rutile-type RuO_2 . The low room-temperature resistivity and the temperature dependency show that the layered $\text{K}_{0.2}\text{RuO}_{2.1}$ exhibits metallic conduction, equivalent to the rutile-type RuO_2 .

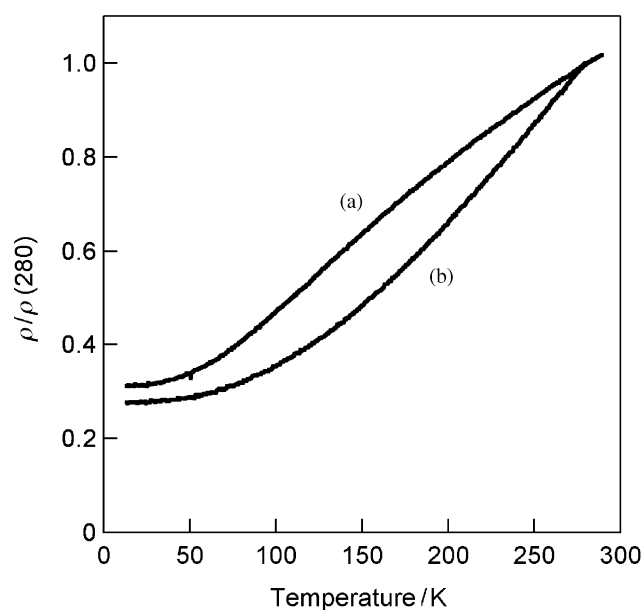


Fig. 1. The normalized resistivity as a function of the temperature for sintered pellets of: (a) layered $\text{K}_{0.2}\text{RuO}_{2.1}$ and (b) rutile-type RuO_2 .

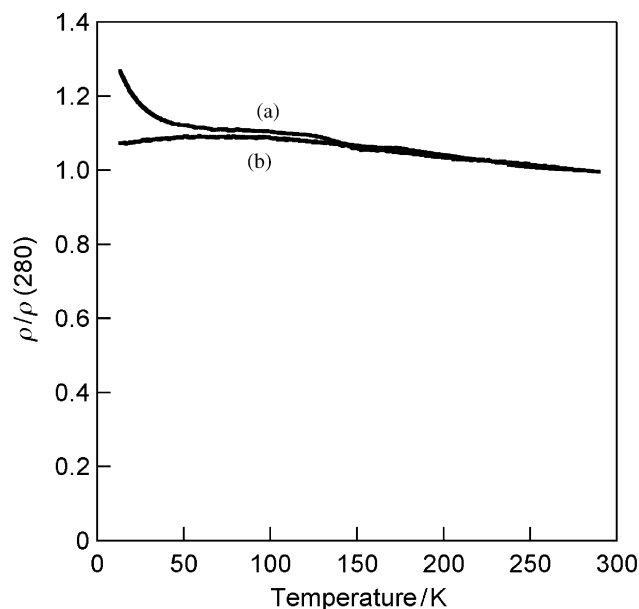


Fig. 2. The normalized resistivity as a function of the temperature for non-sintered pellets of: (a) layered $\text{H}_{0.2}\text{RuO}_{2.1}$ and (b) $\text{RuO}_2 \cdot n\text{H}_2\text{O}$.

Fig. 2 shows the resistivity behavior of non-sintered pellets of layered $\text{H}_{0.2}\text{RuO}_{2.1}$ and amorphous $\text{RuO}_2 \cdot n\text{H}_2\text{O}$. The resistivity behavior of layered $\text{H}_{0.2}\text{RuO}_{2.1}$ is comparable with amorphous $\text{RuO}_2 \cdot n\text{H}_2\text{O}$, with the resistivity slightly decreasing with increasing temperature. The room-temperature resistivity of layered $\text{H}_{0.2}\text{RuO}_{2.1}$ and amorphous $\text{RuO}_2 \cdot n\text{H}_2\text{O}$ was 21 and 1 $\text{m}\Omega\text{cm}$, respectively. Although the temperature dependency of the resistivity of layered $\text{H}_{0.2}\text{RuO}_{2.1}$ and amorphous $\text{RuO}_2 \cdot n\text{H}_2\text{O}$ is not typical of metallic behavior, the temperature dependency is very weak, which suggests that the conduction mechanism probably does not involve a thermally activated process. The non-sintered state should lead to a relatively high interparticle resistance. The low room-temperature resistivity in the order of $\text{m}\Omega\text{cm}$ and weak temperature dependency despite the non-sintered state of layered $\text{H}_{0.2}\text{RuO}_{2.1}$ and amorphous $\text{RuO}_2 \cdot n\text{H}_2\text{O}$ implies that these materials are highly electro-conductive.

In order to gain further insight into the properties of the layered ruthenates, magnetic susceptibility measurements were conducted on powder samples. The magnetic susceptibility χ of $M_{0.2}\text{RuO}_{2.1}$ ($M = \text{K}^+$, H^+ , $\text{C}_2\text{H}_5\text{NH}_3^+$, $(\text{C}_4\text{H}_9)_4\text{N}^+$) is shown as a function of the temperature in Fig. 3. The magnetic susceptibility χ is weakly temperature dependent for all the samples between 150 and 300 K. The small magnetic susceptibility of $\sim 100 \times 10^{-6} \text{emu mol}^{-1}$ in this temperature region is characteristic of Pauli paramagnetism. In the case of $\text{K}_{0.2}\text{RuO}_{2.1}$, a minimum in the magnetic susceptibility χ was observed at 120 K, characteristic of antiferromagnetic interactions in two-dimensional sys-

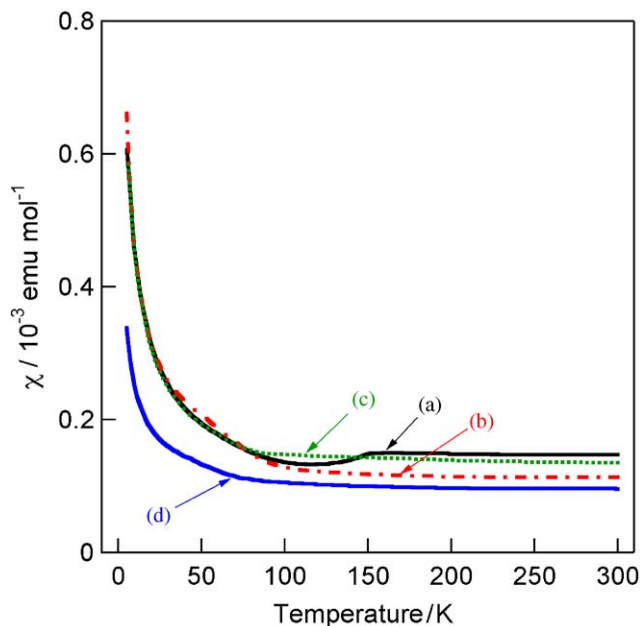


Fig. 3. The magnetic susceptibility for layered: (a) $\text{K}_{0.2}\text{RuO}_{2.1}$, (b) $\text{H}_{0.2}\text{RuO}_{2.1}$, (c) $(\text{C}_2\text{H}_5\text{NH}_3)_{0.2}\text{RuO}_{2.1}$, and (d) $((\text{C}_4\text{H}_9)_4\text{N})_{0.2}\text{RuO}_{2.1}$.

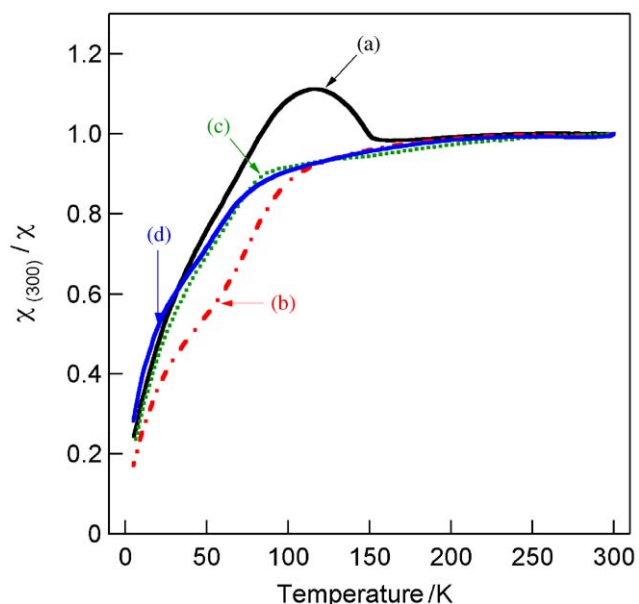


Fig. 4. The reciprocal of the susceptibility normalized to the susceptibility at 300 K ($\chi_{(300)}/\chi$) as a function of temperature for layered: (a) $\text{K}_{0.2}\text{RuO}_{2.1}$, (b) $\text{H}_{0.2}\text{RuO}_{2.1}$, (c) $(\text{C}_2\text{H}_5\text{NH}_3)_{0.2}\text{RuO}_{2.1}$, and (d) $((\text{C}_4\text{H}_9)_4\text{N})_{0.2}\text{RuO}_{2.1}$.

tems [25–27]. The temperature dependency of the magnetic susceptibility of $M_{0.2}\text{RuO}_{2.1}$ ($M = \text{K}^+$, H^+ , $\text{C}_2\text{H}_5\text{NH}_3^+$, $(\text{C}_4\text{H}_9)_4\text{N}^+$) is similar to that of NaRuO_2 and $\text{Na}_x\text{RuO}_2 \cdot n\text{H}_2\text{O}$ [23].

Fig. 4 shows the reciprocal of the susceptibility normalized to the susceptibility at 300 K ($\chi_{(300)}/\chi$) as a function of temperature. The temperature dependency

of the magnetic susceptibility for layered $\text{H}_{0.2}\text{RuO}_{2.1}$ is comparable to that of layered $\text{K}_{0.2}\text{RuO}_{2.1}$, except for the antiferromagnetic transition near 120 K. The similarity in the magnetic behavior of $M_{0.2}\text{RuO}_{2.1}$ ($M = \text{H}^+$, $\text{C}_2\text{H}_5\text{NH}_3^+$, $(\text{C}_4\text{H}_9)_4\text{N}^+$) with metallic $\text{K}_{0.2}\text{RuO}_{2.1}$ suggests that the metallic characteristic of the ruthenate sheets is preserved regardless of the type of interlayer cation (organic or inorganic) and the different basal spacing. The apparent non-metallic behavior of layered $\text{H}_{0.2}\text{RuO}_{2.1}$ observed by the resistivity measurements (Fig. 2) may be due to inter-particle resistance, rather than an intrinsic characteristic of the differences in the interlayer cation. Hence, under the present experimental conditions, the electric and magnetic properties of the ion-exchangeable layered ruthenate seems to be dominated by the ruthenate sheets and the magnetic interaction between the sheets seems to be small. To the best of our knowledge, the layered ruthenate $M_{0.2}\text{RuO}_{2.1}$ is the first example of a family of highly conducting, chemically and electrochemically stable layered oxide that can easily be chemically modified through conventional intercalation chemistry.

4. Conclusions

A series of layered ruthenates, $M_{0.2}\text{RuO}_{2.1}$ ($M = \text{K}^+$, H^+ , $\text{C}_2\text{H}_5\text{NH}_3^+$, $(\text{C}_4\text{H}_9)_4\text{N}^+$), was prepared by a solid-state reaction, an ion-exchange reaction, and a guest-exchange reaction. The resistivity data for sintered polycrystalline $\text{K}_{0.2}\text{RuO}_{2.1}$ revealed metallic conduction between 300 and 13 K. The magnetic susceptibility data of powder samples revealed that $M_{0.2}\text{RuO}_{2.1}$ ($M = \text{K}^+$, H^+ , $\text{C}_2\text{H}_5\text{NH}_3^+$, $(\text{C}_4\text{H}_9)_4\text{N}^+$) shows similar behavior, despite the differences in the interlayer cation. The results suggest that the ruthenate sheets in the ion-exchangeable layered ruthenates possess metallic behavior, making it the first example of a highly conducting, chemically and electrochemically stable layered oxide that can be easily chemically modified through conventional intercalation chemistry.

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